

## Rechargeable Li–O<sub>2</sub> batteries with a covalently coupled MnCo<sub>2</sub>O<sub>4</sub>–graphene hybrid as an oxygen cathode catalyst†

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We employ a MnCo<sub>2</sub>O<sub>4</sub>–graphene hybrid material as the cathode catalyst for Li–O<sub>2</sub> batteries with a non-aqueous electrolyte. The hybrid is synthesized by direct nucleation and growth of MnCo<sub>2</sub>O<sub>4</sub> nanoparticles on reduced graphene oxide, which controls the morphology, size and distribution of the oxide nanoparticles and renders strong covalent coupling between the oxide nanoparticles and the electrically conducting graphene substrate. The inherited excellent catalytic activity of the hybrid leads to lower overpotentials and longer cycle lives of Li–O<sub>2</sub> cells than other catalysts including noble metals such as platinum. We also study the relationships between the charging–discharging performance of Li–O<sub>2</sub> cells and the oxygen reduction and oxygen evolution activity of catalysts in both aqueous and non-aqueous solutions.

Lithium ion batteries (LIBs) have become the main power source for today's portable electronics and are being actively pursued for propelling electric vehicles in the near future.<sup>1–4</sup> However, challenges

remain for LIBs to become a major energy supply device for transportation. In particular, the energy density of LIBs should be increased by at least 3 times in order to support a driving range of more than 500 km with a single charge.<sup>5</sup> Also, the cost of LIBs should be lowered in order to be competitive with respect to other sources of energy. Limited by the insufficient capacity of electrode materials, the current LIB systems are not likely to reach the specific energy level needed for electric transportation in the long run.<sup>1–5</sup> It is necessary to develop alternative types of batteries that are capable of delivering higher energy density.

Li–O<sub>2</sub> batteries have recently attracted renewed interest due to significantly higher gravimetric energy density than LIBs.<sup>5–7</sup> With a theoretical specific energy of ~3500 W h kg<sup>–1</sup> based on the mass of Li and O<sub>2</sub>, Li–O<sub>2</sub> batteries are estimated to provide ~500 to 900 W h kg<sup>–1</sup> in practical devices, which is more than 3 times higher than that of a typical LIB.<sup>5–7</sup> In an ideal Li–O<sub>2</sub> cell, a Li metal anode is oxidized while O<sub>2</sub> is reduced at the cathode during discharging, producing Li<sub>2</sub>O<sub>2</sub> in an aprotic electrolyte or LiOH in an alkaline solution. During charging, Li<sub>2</sub>O<sub>2</sub> or LiOH is supposed to be oxidized to generate O<sub>2</sub> at the cathode and Li is plated back onto the anode. Despite that numerous cathode catalysts including carbon, metal oxides and noble metals have been applied to enhance the sluggish kinetics of the oxygen reduction reaction (ORR) and oxygen evolution reaction (OER) at the cathode, Li–O<sub>2</sub> cells obtained thus far have exhibited high overpotential and short cycle lives.<sup>5–12</sup> It is important to understand the operating principles of Li–O<sub>2</sub> cells and to explore new oxygen electrode materials with higher catalytic

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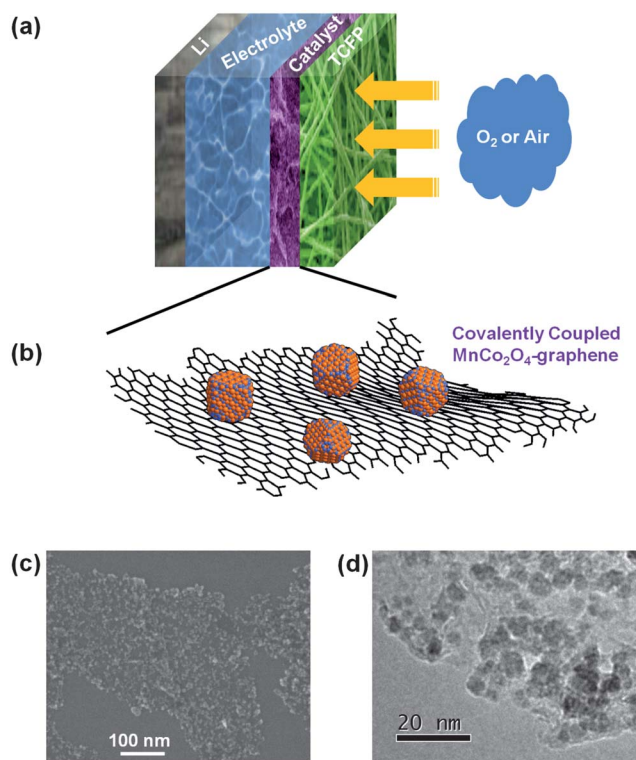
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### Broader context

A developing trend in the automobile industry now is to build electric vehicles that do not rely on fossil fuels and have zero emission. To provide electric vehicles with a driving range comparable to that of vehicles powered by gasoline or diesel, batteries with several times higher energy density than the existing Li ion batteries are needed. The Li–O<sub>2</sub> battery is such a choice. However, Li–O<sub>2</sub> cells obtained thus far have exhibited low round trip efficiency and short cycle lives mainly due to the sluggish kinetics of oxygen reduction and evolution reactions at the cathode. It is highly important to understand the scientific principles as well as to explore new oxygen electrode materials with higher catalytic activity for Li–O<sub>2</sub> cells. In this work, we design and fabricate Li–O<sub>2</sub> coin cells with a covalently coupled MnCo<sub>2</sub>O<sub>4</sub>–graphene hybrid material as a catalyst for the O<sub>2</sub> cathode. The cells show lower overpotentials and longer cycle lives than the ones catalyzed by other catalysts. We also study the relationships between the charging–discharging performance of Li–O<sub>2</sub> cells and the oxygen reduction/evolution activity of catalysts.



**Fig. 1**  $\text{MnCo}_2\text{O}_4$ -graphene hybrid as a cathode catalyst for  $\text{Li-O}_2$  batteries. (a) Schematic structure of the  $\text{Li-O}_2$  cell catalyzed by the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid. (b) Schematic structure of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid material comprised of  $\text{MnCo}_2\text{O}_4$  nanoparticles covalently bonded to NGO sheets through carbon-oxygen-metal and carbon-nitrogen-metal bonds.<sup>18</sup> (c) An SEM image of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid. (d) A TEM image of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid.

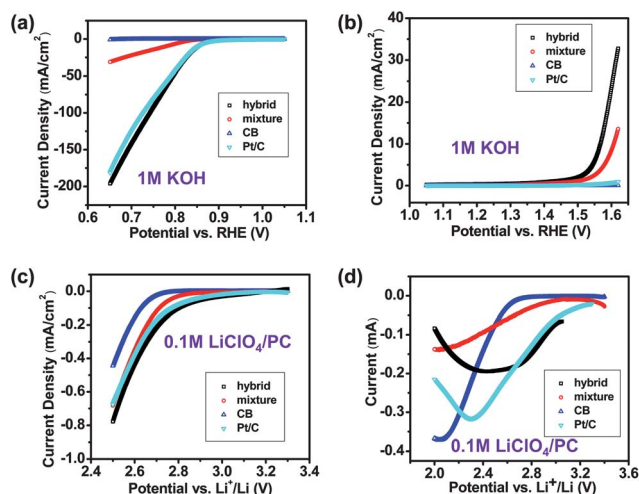
activity to enhance the performance of  $\text{Li-O}_2$  cells by lowering the overpotential and improving cycle life.

Transition metal oxides such as  $\text{Co}_3\text{O}_4$  and  $\text{MnO}_2$  have been reported to be catalytically active for  $\text{Li-O}_2$  cells.<sup>8,13-16</sup> Recently, by growing  $\text{Co}_3\text{O}_4$  nanoparticles on N doped mildly oxidized graphene sheets, we have synthesized a hybrid catalyst with excellent bi-functional activity for ORR and OER in aqueous alkaline solutions.<sup>17</sup> Through substitution of  $\text{Mn}^{3+}$  for  $\text{Co}^{3+}$  in the spinel lattice, ORR catalytic performance of the resulting  $\text{Co(II)Co(III)Mn(III)O}_4$ -graphene hybrid catalyst was further improved.<sup>18</sup> Covalent coupling between the oxide nanoparticles and graphene led to drastically improved catalytic activity of the hybrid over a physical mixture of the two by conventional means.<sup>18</sup> In this work we utilize the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid material as a cathode catalyst for  $\text{Li-O}_2$  cells (Fig. 1) and explore whether the ORR and OER electrocatalytic activities in aqueous solutions can be translated to organic electrolytes used for  $\text{Li-O}_2$  batteries. At a current density of  $100 \text{ mA g}^{-1}$ , our  $\text{Li-O}_2$  cell operating with a carbonate electrolyte exhibits a discharging voltage of  $\sim 2.95 \text{ V}$  and a charging voltage of  $\sim 3.75 \text{ V}$ , among the lowest overpotentials (similar to a Pt/C catalyst) reported in a similar electrolyte at comparable gravimetric current densities. The  $\text{Li-O}_2$  cell with the  $\text{MnCo}_2\text{O}_4$ -graphene cathode catalyst also shows better charge-discharge cycling stability than that with a Pt/C catalyst. A capacity of  $1000 \text{ mA h g}^{-1}$  can be delivered for 40 cycles without significant increase in overpotential. The high cell performance is

attributed to the excellent electrocatalytic activity of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid catalyst. The discharging performance of the  $\text{Li-O}_2$  cell correlates well with the high ORR activity of our  $\text{MnCo}_2\text{O}_4$ -graphene hybrid catalyst in aqueous and non-aqueous solutions. However, we find that the OER catalytic activities of various catalysts in aqueous solutions do not correlate with the charging performance of relating  $\text{Li-O}_2$  cells when comparing our hybrid catalyst with other materials including Pt/C.

The  $\text{MnCo}_2\text{O}_4$ -graphene hybrid was synthesized by a two step solution phase method in which  $\text{Co(OAc)}_2$  and  $\text{Mn(OAc)}_2$  were first hydrolyzed and coated onto graphene oxide (GO) in an ethanol-water mixed solvent with the addition of ammonium hydroxide. The suspension was then solvothermally treated to produce the hybrid material.<sup>17,18</sup> We showed recently that strong chemical coupling between  $\text{Co}_3\text{O}_4$  nanoparticles with N doped reduced GO (NGO) substrates through covalent bonding led to excellent ORR and OER bi-functional catalytic activities in aqueous KOH solutions (Fig. 2a and b).<sup>17</sup> The  $\text{MnCo}_2\text{O}_4$ -graphene hybrid showed an enhanced ORR activity over the  $\text{Co}_3\text{O}_4$ -graphene hybrid due to an increase in the ORR active sites through  $\text{Mn}^{3+}$  substitution for  $\text{Co}^{3+}$  and an increase in the electrochemically active surface area.<sup>18</sup> The hybrid material showed much enhanced ORR and OER activities over free oxide nanoparticles physically mixed with conductive carbon (Fig. 2a and b).

We assessed the ORR catalytic activity of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid in a non-aqueous electrolyte of  $\text{O}_2$ -saturated  $0.1 \text{ M LiClO}_4$  in propylene carbonate (PC) together with Pt/C, carbon black and a physical mixture of  $\text{MnCo}_2\text{O}_4$  and NGO for comparison. Loaded onto Teflon coated carbon fiber papers (TCFPs), the catalysts exhibited a similar trend of relative activities as in aqueous media



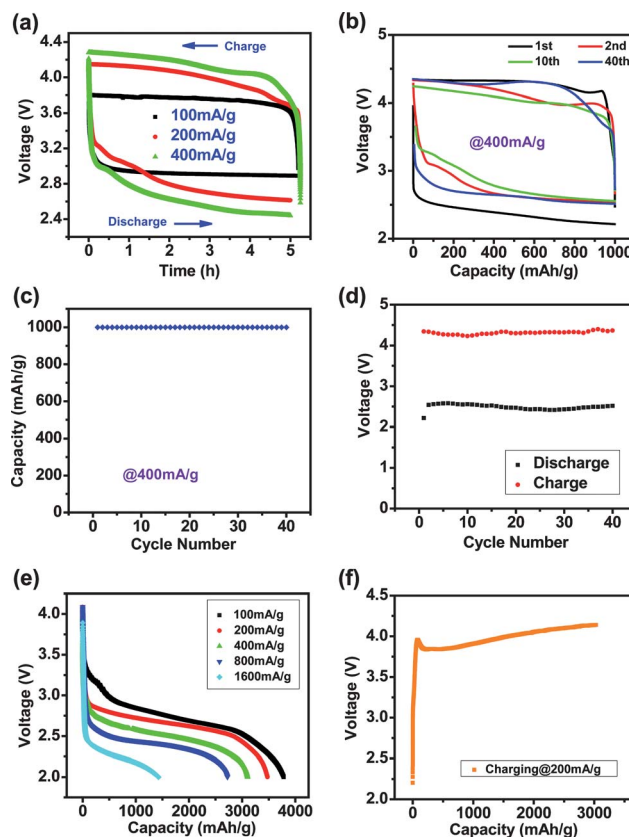
**Fig. 2** ORR and OER with the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid electrocatalyst in aqueous and non-aqueous media. (a) ORR catalytic activity of the hybrid compared to control catalysts in  $1 \text{ M KOH}$  solution. The mixture was made by physically mixing the free  $\text{MnCo}_2\text{O}_4$  nanoparticles with NGO. CB stands for carbon black (Super P from Timcal). Pt/C is a commercial catalyst with  $20 \text{ wt}\%$  Pt on Vulcan carbon black (from Fuel Cell Store). Catalysts were loaded on TCFP. (b) OER catalytic activity of the hybrid compared to control catalysts in  $1 \text{ M KOH}$  solution. Catalysts were loaded on TCFP. (c and d) ORR catalytic activity of the hybrid compared to control catalysts in  $0.1 \text{ M LiClO}_4/\text{PC}$  solution. Catalysts were loaded on TCFP (c) and glassy carbon electrode (d) respectively.

(Fig. 2a and c). At a potential of 2.8 V vs. Li<sup>+</sup>/Li, the ORR current (corrected with current measured in an Ar-saturated electrolyte) of the hybrid electrode ( $\sim 0.10 \text{ mA cm}^{-2}$ ) was slightly higher than that of the Pt/C electrode ( $\sim 0.08 \text{ mA cm}^{-2}$ ), and much higher than those of the physical MnCo<sub>2</sub>O<sub>4</sub> and NGO mixture ( $\sim 0.03 \text{ mA cm}^{-2}$ ) and carbon black ( $\sim 0.001 \text{ mA cm}^{-2}$ ). ORR measurements with catalysts loaded onto glassy carbon electrodes also confirmed the trend in relative activity of the catalysts (Fig. 2d). The MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid and Pt/C exhibited similar ORR onset potentials  $>3.0 \text{ V vs. Li}^+/\text{Li}$ , considerably higher than those of the physical MnCo<sub>2</sub>O<sub>4</sub> and NGO mixture ( $\sim 3.0 \text{ V}$ ) and carbon black ( $\sim 2.8 \text{ V}$ ). The electrochemical results suggested that our MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid could be an effective ORR catalyst for Li-O<sub>2</sub> batteries.

We made a Li-O<sub>2</sub> cell cathode by loading  $\sim 0.5 \text{ mg}$  of the MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid material onto a  $\sim 1 \text{ cm}^2$  TCFP (Fig. S1a†). TCFP was widely used as a catalyst support and current collector in fuel cells.<sup>17,18</sup> We found it useful for preparing the oxygen electrode in Li-O<sub>2</sub> cells as the TCFP surface was not wetted by the electrolyte so that the gas could diffuse to the catalyst easily through the porous structure of TCFP. Our Li-O<sub>2</sub> cell was assembled with standard 2032 type coin cell cases except for the holes drilled in the positive side of the cases for oxygen intake (Fig. S1a and b†). Oxygen was prevented from reaching the anode by the electrolyte layer. It was a simple and efficient method to fabricate Li-O<sub>2</sub> cells utilizing commercial coin cell parts with minor modifications. 1 M LiClO<sub>4</sub> in PC was used as the electrolyte and Li metal served as the anode. The Li-O<sub>2</sub> cells were tested in a dry box with O<sub>2</sub> or air flow (Fig. S1c†).

Typical charge and discharge voltage profiles of our Li-O<sub>2</sub> cell are shown in Fig. 3a. At a current density of  $100 \text{ mA g}^{-1}$  based on the total mass of the MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid material, the average discharging voltage was about 2.95 V, close to the thermodynamic potential of the reaction  $2\text{Li}^+ + 2\text{e}^- + \text{O}_2 \rightarrow \text{Li}_2\text{O}_2$ , while the average charging voltage was about 3.75 V,  $\sim 0.8 \text{ V}$  higher than the discharging voltage. This overpotential was comparable to that of precious metal based catalysts such as Pt/Au nanoparticles<sup>9</sup> and Pt/C,<sup>19</sup> and lower than those of other reported catalysts such as graphene,<sup>10,20</sup> metal oxides,<sup>8,21</sup> lithium metal oxides,<sup>22</sup> Pd-MnO<sub>2</sub>,<sup>14</sup> MoN-graphene<sup>11</sup> and CoMn<sub>2</sub>O<sub>4</sub>-graphene,<sup>16</sup> at comparable current densities in similar carbonate electrolytes. The overpotential increased at higher current densities (Fig. 3a), but the charging potential at  $400 \text{ mA g}^{-1}$  was still below the high voltage limit of the electrolyte.<sup>2,23</sup> Cycle life of the Li-O<sub>2</sub> cell was tested with a capacity cut-off of  $1000 \text{ mA h g}^{-1}$  at a current density of  $400 \text{ mA g}^{-1}$ . The cell showed good cycling ability over 40 cycles in dry oxygen (Fig. 3c). During cycling, the final voltage of each discharge segment stabilized at  $\sim 2.5$  to  $2.6 \text{ V}$  except for the first discharge, and the final voltage of each charging segment was in the range of  $\sim 4.2$  to  $4.3 \text{ V}$  (Fig. 3b and d). This was among the best cycling performance of Li-O<sub>2</sub> cells operating in carbonate electrolytes.<sup>8-16,20-22</sup>

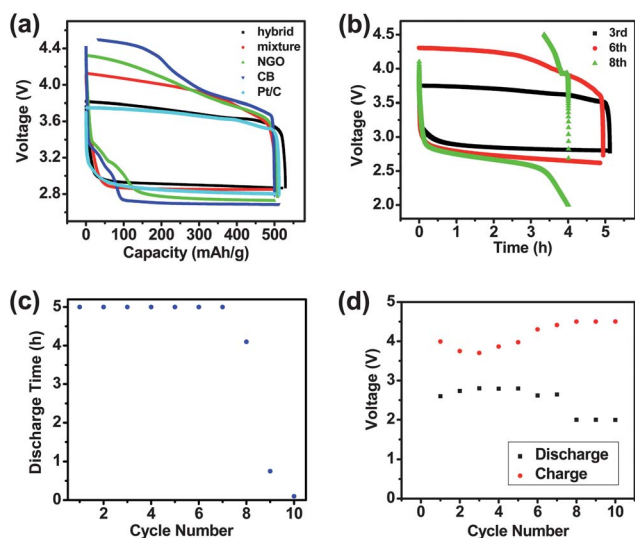
We further examined the discharge power rate of the Li-O<sub>2</sub> cell catalyzed by the MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid. Upon charging at a fixed current density of  $200 \text{ mA g}^{-1}$  for 15 h (Fig. 3f), the Li-O<sub>2</sub> cell was discharged to 2.0 V at various current densities (Fig. 3e). At  $100 \text{ mA g}^{-1}$ , a specific discharge capacity of  $\sim 3784 \text{ mA h g}^{-1}$  was obtained based on the total mass of the MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid, which remained as  $\sim 2743 \text{ mA h g}^{-1}$  at a high discharge current density of  $800 \text{ mA g}^{-1}$  (Fig. 3e), indicating high capacity and good rate capability of the cell.<sup>8,11,13-16,20,21</sup> This suggests that our MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid material holds promise in catalyzing Li-O<sub>2</sub> cells with high energy density.



**Fig. 3** A MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid catalyzed Li-O<sub>2</sub> cell. (a) Charging and discharging voltage profiles of the cell at various current densities. (b) Charging and discharging voltage profiles of the cell at various cycle numbers at a current density of  $400 \text{ mA g}^{-1}$ . (c) Specific discharge capacity of the cell over 40 cycles at  $400 \text{ mA g}^{-1}$ . (d) Cell voltage upon completion of each discharge (black) and charge (red) segment over the 40 cycles in (c). (e) Discharging voltage profiles of the cell at various current densities. (f) A typical charging voltage profile of the cell for power rate measurements in (e).

Since a practical Li-O<sub>2</sub> cell will utilize O<sub>2</sub> in ambient air instead of pure oxygen gas, we tested our Li-O<sub>2</sub> cells in dry air environment. A charge-discharge profile is plotted in Fig. S2a†. In dry air, the cell showed an average discharging voltage of  $\sim 2.85 \text{ V}$  (Fig. S2a†), slightly lower than that in pure oxygen due to lower O<sub>2</sub> concentration in air. The average charging voltage was  $\sim 3.80 \text{ V}$ , similar to that in pure oxygen. Our MnCo<sub>2</sub>O<sub>4</sub>-graphene catalyzed Li-O<sub>2</sub> cell also demonstrated good rechargeability in dry air. It was able to deliver a specific capacity of  $1000 \text{ mA h g}^{-1}$  for more than 20 cycles (Fig. S2b†).

In order to compare the catalytic properties of the MnCo<sub>2</sub>O<sub>4</sub>-graphene hybrid, we prepared Li-O<sub>2</sub> cells with other catalyst materials typically used in oxygen electrochemistry including N-doped graphene, metal oxide nanoparticles mixed with conducting carbon black, and Pt/C. The cells were discharged and charged at  $100 \text{ mA g}^{-1}$  with a capacity cut-off of  $500 \text{ mA h g}^{-1}$  and a high voltage cut-off of 4.5 V. Among these catalysts, carbon black showed the highest overpotential. Although discharging could be steadily performed at  $\sim 2.70 \text{ V}$  (Fig. 4a, blue curve), the cell with carbon black as the cathode catalyst was barely rechargeable. The voltage reached the 4.5 V cut-off before the charging completed even at a low current



**Fig. 4** Comparison of  $\text{MnCo}_2\text{O}_4$ -graphene hybrid material to other cathode catalysts in  $\text{Li-O}_2$  cells. (a) Charging and discharging voltage profiles of  $\text{Li-O}_2$  cells catalyzed by different catalysts at a current density of  $100 \text{ mA g}^{-1}$ . NGO was made in the same way as the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid but without any metal precursors added (ref. 18 in text). (b) Charging and discharging voltage profiles of the  $\text{Li-O}_2$  cell catalyzed by Pt/C at various cycle numbers. (c) Specific discharge capacity of the  $\text{Li-O}_2$  cell catalyzed by Pt/C over 10 cycles. Current density was  $100 \text{ mA g}^{-1}$  for the first 6 cycles and  $200 \text{ mA g}^{-1}$  for the rest of cycles. (d) Cell voltage upon completion of each discharge (black) and charge (red) segment over the 10 cycles in (c).

density of  $100 \text{ mA g}^{-1}$  (Fig. 4a, green curve). NGO showed a slightly higher discharging voltage than carbon black as NGO with N doping was expected to be more active in ORR than non-doped carbon materials.<sup>17,24</sup> The charging voltage of the NGO catalyzed cell was lower than that of the carbon black cell. However the voltage still reached as high as 4.3 V at the end of the charging process, indicating poor catalytic activity of NGO for the charging reaction.

Physically mixing NGO with free  $\text{MnCo}_2\text{O}_4$  nanoparticles reduced both the charging and discharging overpotentials considerably. At  $100 \text{ mA g}^{-1}$ , the  $\text{Li-O}_2$  cell catalyzed by the mixture showed a discharging voltage of  $\sim 2.85 \text{ V}$  and the charging completed at a voltage of  $\sim 4.10 \text{ V}$  (Fig. 4a, red curve). However, the charging and discharging overpotentials of the mixture catalyzed  $\text{Li-O}_2$  cell were still substantially higher than those of the cell catalyzed by the covalently coupled  $\text{MnCo}_2\text{O}_4$ -graphene hybrid (Fig. 4a, black curve).

We found that Pt/C was a similarly active catalyst to the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid, significantly outperforming the free metal oxide nanoparticles mixed with conductive carbon (Fig. 4a, cyan curve). The high activity of Pt/C for  $\text{Li-O}_2$  cells agrees with previous studies.<sup>9,19</sup> Despite relatively low overpotential in the initial cycles, the Pt/C catalyzed  $\text{Li-O}_2$  cell showed a much faster increase in overpotential and shorter cycle life than the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid catalyzed cell (Fig. 4b–d). The Pt/C cell was first cycled at  $100 \text{ mA g}^{-1}$  with a capacity cut-off of  $500 \text{ mA h g}^{-1}$  for 6 cycles. From the 3<sup>rd</sup> to the 6<sup>th</sup> cycle, the discharging voltage decreased to  $\sim 2.70 \text{ V}$  from  $\sim 2.85 \text{ V}$ , while the charging voltage increased to  $\sim 4.30 \text{ V}$  from  $\sim 3.70 \text{ V}$  (Fig. 4b and d). Current density was increased to  $200 \text{ mA g}^{-1}$  starting from the 7<sup>th</sup> cycle and a substantial decay in capacity was observed in the 8<sup>th</sup> cycle (Fig. 4b and c). The capacity of the Pt/C cell

further dropped to zero in the 10<sup>th</sup> cycle (Fig. 4c). After 10 discharge-charge cycles, SEM imaging revealed that the Pt/C catalyst was buried in a thick layer of coating material (Fig. S3a and b†) likely resulted from side reactions. The coating appeared to be electrochemically inactive and was not removed during charging. In contrast, the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid catalyst after 10 discharge-charge cycles retained its structure and morphology, without the formation of a thick coating layer (Fig. S3c and d†).

Although the NGO catalyzed  $\text{Li-O}_2$  cell exhibited higher discharge voltage and lower charging voltage than the carbon black cell, there was still a substantially higher overpotential than that of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid cell, especially for the charging reaction. The voltage of the NGO cell was  $\sim 4.30 \text{ V}$  upon completion of charging,  $\sim 0.50 \text{ V}$  higher than that of the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid cell. High recharging overpotential is often a problem associated with carbon material catalyzed  $\text{Li-O}_2$  cells discussed in the literature, despite high specific capacity in the first discharge due to the large surface area of the carbon cathode.<sup>10,12</sup> Free  $\text{MnCo}_2\text{O}_4$  nanoparticles physically mixed with NGO showed improved catalytic activity for  $\text{Li-O}_2$  cells over NGO, comparable to other reported metal oxide nanocrystal catalysts mixed with conductive carbon.<sup>8,21</sup> Importantly, the lowest overpotential was achieved for the  $\text{Li-O}_2$  cell catalyzed by the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid, attributed to the excellent electrocatalytic activity afforded by the strong electrochemical coupling between the graphene sheets and the nanoparticles selectively grown and covalently bonded on graphene. Such intimate and effective interaction within the hybrid structure could result in fast and facile electron transfer through the conducting graphene network, which facilitates the charging and discharging reactions catalyzed on the surface of the oxide nanoparticles. Similar synergistic electrochemical effects have been also observed for various inorganic nanocrystals grown on reduced graphene oxide.<sup>17,18,25–34</sup> Although Pt/C was able to deliver as high catalytic activity as the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid, the Pt/C catalyzed  $\text{Li-O}_2$  cell was much less stable through charge-discharge cycling, caused by the formation of electrochemically inactive coating materials that could block the catalyst surface (Fig. S3†). Therefore the  $\text{MnCo}_2\text{O}_4$ -graphene hybrid could be a potential catalyst with high activity, good cycling stability and low cost for  $\text{Li-O}_2$  batteries.

We found that the performances of various catalysts for the discharging reaction of the  $\text{Li-O}_2$  cells (Fig. 4a) were consistent with the ORR activity trend measured in both non-aqueous (Fig. 2c and d) and aqueous electrolytes (Fig. 2a). However, the catalytic capability for the charging reaction in a non-aqueous electrolyte (Fig. 4a) was not directly correlated with the OER activity in aqueous solutions (Fig. 2b). In particular, Pt/C was barely active for OER in 1 M KOH solution, but was able to catalyze the charging reaction in  $\text{Li-O}_2$  cells with low overpotential (similar to our hybrid which was differed from Pt/C with high OER activity in aqueous KOH solutions). This suggested that charging reactions in the  $\text{Li-O}_2$  cells differed from OER and were not catalyzed by regular OER catalysts for aqueous systems. It has been recently reported that the discharge products of  $\text{Li-O}_2$  cells with carbonate electrolytes are much more complicated than simple  $\text{Li}_2\text{O}_2$ .<sup>35–37</sup> A dominant amount of the solid discharge products could be comprised of lithium alkylcarbonates ( $\text{LiRCO}_3$ ) and lithium carbonate ( $\text{Li}_2\text{CO}_3$ ), resulted from nucleophilic attack on the organic carbonates by discharge intermediates such as superoxide ions ( $\text{O}_2^-$ ) and the resulting decomposition of the carbonate electrolyte.<sup>35–37</sup> Although the carbonate electrolyte is frequently used in

the present Li–O<sub>2</sub> cells,<sup>8–16,20–22</sup> the issue of electrolyte degradation has to be solved before a stable Li–O<sub>2</sub> battery can be developed.<sup>35–37</sup> Moreover, we noticed that there was a significantly higher charging overpotential than that of discharging for all of the catalysts tested (Fig. 4a). The chemistry and catalysis involved in the charging process in Li–O<sub>2</sub> batteries are currently not understood. Such an understanding is needed in order to develop a new generation of catalysts with significantly improved activity for the charging reactions of Li–O<sub>2</sub> batteries.

In summary, we have shown a covalently coupled MnCo<sub>2</sub>O<sub>4</sub>–graphene hybrid material as an active, stable and low-cost cathode catalyst for Li–O<sub>2</sub> batteries. The Li–O<sub>2</sub> cell with the hybrid catalyst has similar low charge–discharge overpotentials as the Pt/C catalyzed cell, but with a much longer cycle life. Owing to electrochemical coupling between the graphene sheets and the MnCo<sub>2</sub>O<sub>4</sub> nanoparticles in the hybrid, the material outperforms other metal oxide based and carbon based catalysts under similar measurement conditions, affording Li–O<sub>2</sub> coin cells with high capacity, low overpotential and good cycling stability. Further work is needed to understand and improve electrocatalysis in the charging reactions in Li–O<sub>2</sub> batteries.

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